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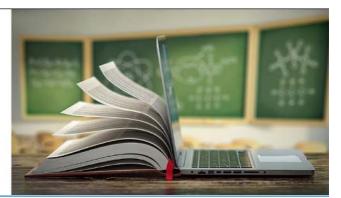
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Long single crystalline  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods: facile synthesis and photocatalytic application

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#### Abstract

Single crystalline cubic sesquioxide bixbyite  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods have been synthesized successfully by a simple, low cost, environmental benign hydrothermal route. As synthesized  $\gamma$ -MnOOH were calcined at 600 °C to obtain  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods, which were further subjected to various characterizations. The alpha manganese sesquioxide cubic bixbyite-type oxide formation was confirmed by the XRD studies. The surface morphology and elemental analysis were explored by SEM with EDX studies, respectively. High-resolution transmission electron microscopy HRTEM and SAED showed that the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods were single crystalline and were grown along the *C*-axis of the crystal plane. The UV–visible spectrum indicated that the absorption was prominent in the ultraviolet region. In addition, PL spectrum result of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods recommended possible photocatalytic applications. The photocatalyst ensures a new platform for the decolorization of dye molecules of the harmful cationic dyes like methylene blue and rhodamine B. Possible growth mechanism and photocatalytic dye degradation mechanism were proposed for synthesized  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods.

# 1. Introduction

Wastewater treatment and recycling are important concerns and researchers are looking forward to developing inexpensive and suitable technology for wastewater treatment. The wastewater has been treated by an advanced oxidation process, which is considered as attractive and effective technology in recent days [1–3]. Photocatalysis is a promising method, which can be used for the degradation of various organic and inorganic pollutants in wastewater [4, 5]. Nanosized semiconductor materials such as  $TiO_2$ , ZnO, CdS, ZnS, and  $WO_3$  and  $Fe_2O_3$  have the ability to play the role of higher photocatalytic activity. The two predominant photocatalytic materials found in the literature are considered as  $TiO_2$  and ZnO because of their wide-bandgap semiconducting properties. But their photocatalytic activity is greatly decreased due to the rapid recombination rate of photoexcited electronhole pairs [6, 7]. Recently, a few manganese oxide-based materials have been developed as alternative materials as a photocatalyst without any further modification. Manganese oxides have been the subject of much research due to the magnetic, electrical, and catalytic properties, structural variability, and different oxidation states and their broad range of physical and optical properties [8–10]. In nature, manganese oxide exhibits different oxidation states, which consists of MnO, MnO<sub>2</sub>, Mn<sub>2</sub>O<sub>3</sub>, Mn<sub>3</sub>O<sub>4</sub>, and Mn<sub>5</sub>O<sub>8</sub> [11]. In particular, the preparation of Mn<sub>2</sub>O<sub>3</sub> has been attracted much research attention because of its environmentally friendly active catalyst for removing CO and NO from various sources.

There a variety of chemical and physical synthetic routes are available to synthesize Mn<sub>2</sub>O<sub>3</sub> nanostructures, including hydrothermal, sol-gel, pulsed laser ablation, and co-precipitation methods [12–14]. The hydrothermal process offers several advantages over other methods such as lower energy consumption, reduced

environmental impact, controlled morphology, and high crystallinity [15]. At present, many works have been devoted to the synthesis and property studies of 1D Mn<sub>2</sub>O<sub>3</sub> nanostructured materials, owing to their anisotropic dimension [16]. Many results in the literature support the existence of a large number of charge carriers on the active site of the nanostructures, which leads to a significant increase in the photocatalytic activity in comparison to nanoparticles in the spherical form [17]. Youcun Chena *et al* [18] Javed *et al* synthesized  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods [19] and nanowires employed by hydrothermal methods, Yong Cai *et al* synthesized  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods [12], Pijun Gong *et al* synthesized  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods [20] Yang Zhang *et al* synthesized  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods by hydrothermal treatment has been reported [21].

Gnanam *et al* synthesized (a-Mn<sub>2</sub>O<sub>3</sub>) nanodumb-bells had achieved the 71% degradation of the Remazol red B dye [22], Electrospun Mn<sub>2</sub>O<sub>3</sub> nanowrinkles prepared by Mengzhu Liu *et al* showed catalytic effects on decomposition of methyl blue dye with H<sub>2</sub>O<sub>2</sub> [1] and Jianhui Zhao *et al* employed a facile fabrication of novel Mn<sub>2</sub>O<sub>3</sub> which degraded of ciprofloxacin at 94.7% [4]. Seldom research reports were published on the photocatalytic dye degradation for two cationic dye at different photocatalyst dosage for  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. In addition, the formation and growth mechanism of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods was also studied.

#### 2. Experimental section

#### 2.1. Materials

All chemical components used for the preparation of  $\gamma$ -MnOOH and  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods, used as without any further purification. Potassium permanganate [KMnO<sub>4</sub>], and polyethylene glycol (PEG-400) were purchased from MERCK. Distilled water (DW) was used for the preparation of the aqueous solution. (PEG-400) non-ionic surfactant, low-cost, low toxicity, water-soluble organic polymer, and controls the size, and prevents agglomeration.

#### 2.2. Synthesis of $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods

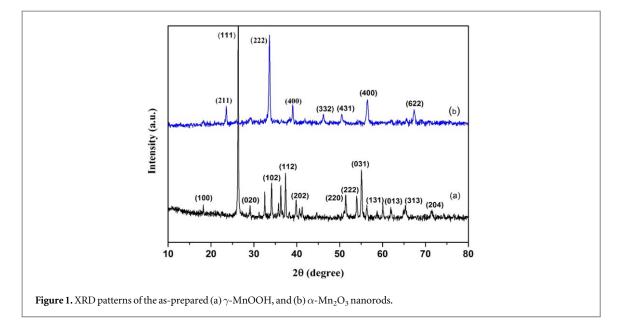
The  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods were prepared by hydrothermal method. 0.5 g of KMnO4 was dissolved in 70 ml of distilled water under magnetic stirring, and 1.9 ml of (PEG-400) was added to the above solution. The purple colour of the solution slowly changed to brown (indicating onset of the formation of manganese oxhydroxide). The manganese oxyhydroxide formed based on the redox reaction after that; the solution was transferred to a 100 ml capacity of Teflon-lined stainless steel autoclave, then hydrothermally treated at 150 °C for 15 h and then cooled to room temperature naturally. The resultant brown solid product was filtered, and vacuum dried at 90 °C for at 12 h. Finally, a brown powder ( $\gamma$ -MnOOH) was obtained that was calcined at 600 °C to form ( $\alpha$ -Mn<sub>2</sub>O<sub>3</sub>) nanorods.

#### 2.3. Characterization

The prepared ( $\alpha$ -Mn<sub>2</sub>O<sub>3</sub>) nanorods were characterized by x-ray powder diffractometer ((XRD), which was obtained using Bruker, D8 Advance. FT-IR data of the product was collected using Fourier transform infrared (FT-IR) spectroscopy that was carried out on IR Affinity-1. The UV data was collected by UV–visible absorbance spectroscopy, and that was conducted using JASCO (V-670 PC). The room temperature of Photo-luminescence spectroscopy (RTPL), which was performed using F-7000 FL spectrophotometer. Morphology studies and chemical compositions were observed through scanning electron microscopy with energy-dispersive x-ray spectroscopy and elemental mapping (Zeiss EVO 18, Germany). High-resolution Transmission Electron Microscopy (HRTEM) was performed using FEI-TECNAIG2-20 TWIN at an operating voltage of 100kv and Bruker EDX with LN2 free detect. Photocatalytic dye degradation application was carried out by Mercury UV light source.

#### 2.4. Photocatalytic measurements

The decolorization of dye molecules reaction was performed by Sankyo Denki Twelve numbers of 8W Mercury UV lamps of 254–365 nm wavelength. The change in the absorbance spectra of Methylene Blue (MB) and Rhodamine B (RB) with various irradiation times were determined using ultraviolet (UV) light source Typically, 6.5 mg of photocatalyst ( $\alpha$ -Mn<sub>2</sub>O<sub>3</sub>) were added to the 100 ml of the aqueous solution of the MB, and 4.5 mg of photocatalyst added to the 100 ml of RB dye. Prior to the irradiation, the suspension was magnetically stirred in the dark for 30 min afterward; the suspensions were irradiated by a UV light source. At given irradiation time, the photo-reacted suspension (3 ml) was taken, and the suspension was analyzed by using a UV-vis spectrophotometer (MODEL: Specord 201 plus).



# 3. Result and discussion

#### 3.1. Structural analysis

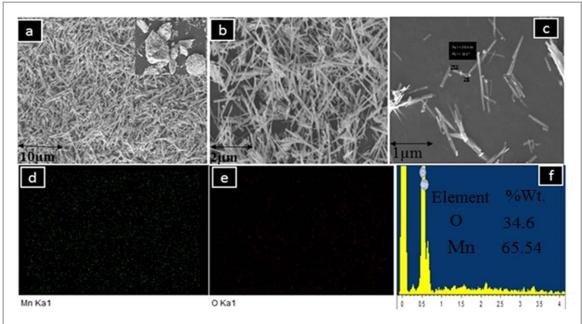
The XRD diffraction patterns of as-prepared  $\gamma$ -MnOOH and calcined  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods are in figure 1. All the diffraction peaks can be indexed to the monoclinic phase of  $\gamma$ -MnOOH figure 1(a). The strong and sharp diffraction patterns can be compared with a standard value of JCPDS file no 41-1379. Figure 1(b) the XRD pattern of the calcined (at 600 °C) sample confirms the bixbyite cubic phase of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. The peaks observed at  $2\theta = 23.2^{\circ}$ ,  $32.9^{\circ}$ ,  $38.2^{\circ}$ ,  $45.3^{\circ}$ ,  $49.3^{\circ}$ ,  $55.1^{\circ}$  and  $65.7^{\circ}$ , which can be attributed to the (211), (222), (400), (332), (431), (440) and (622) plans for  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. (JCPDS 41-1442) [23–25]. There are no other diffraction peaks corresponding to impurities, indicating the high purity of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. The average crystallite size D in nm was estimated using the Debye–Scherrer formula, which is D = K  $\lambda/\beta \cos(\theta)$  where  $\lambda$  is the wavelength of the incident X-ray beam (1.5406 Å),  $\beta$  is the full width at half maximum (FWHM in radian) of the diffraction peak,  $\theta$  is the Bragg diffraction angle, and k is Scherrer constant. The estimated average crystallite size of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods was 31 nm. The lattice parameter of the cubic phase was calculated by using the combined formula of Bragg and the interplanar distance of the cubic structure a = 0.9423 nm.

$$\lambda = \frac{\sqrt{\mathbf{h}^2 + \mathbf{k}^2 + \mathbf{l}^2}}{2\sin\theta} \tag{1}$$

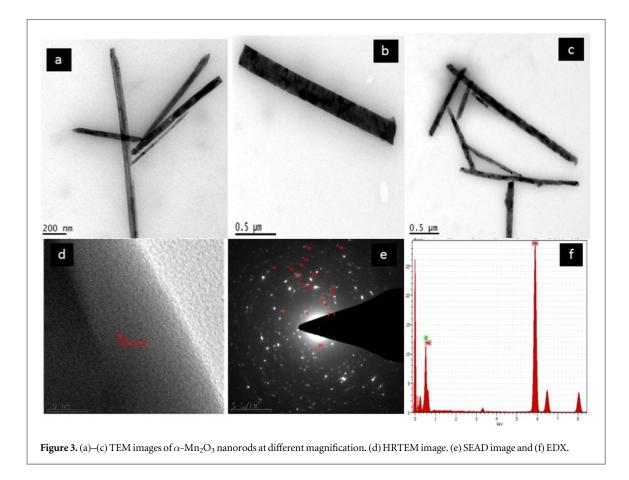
The x-ray density for a cubical system can be calculated using the Dx = z M/NV formula, where z is the number of atoms per unit cell, M is the molecular weight, N is Avogadro's constant and V is the volume of the unit cell. The x-ray density calculated value was (Dx) is 5.0131 g cm<sup>-3</sup> for  $\alpha$ - Mn<sub>2</sub>O<sub>3</sub> nanorods. Further, the specific surface area (SSA) of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods has been calculated with S<sub>a</sub> = 6/D Dx formula [23, 26]. The estimated value of SSA is 38.60 × 10<sup>4</sup> cm<sup>2</sup> g<sup>-1</sup> for  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods.

#### 3.2. Surface morphology analysis

Figures 2(a)–(c) shows the morphologies of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods at different magnifications. The samples clearly reveal the presence of a large quantity of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods aligns in the random orientation. Figures 2(d), (e) shows the elemental mapping of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. Figure 2(f) shows the EDX pattern, which designates the purity of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods by indicative of only Mn and O. The stoichiometrical atomic percentage of Mn is 36% and O is 64%. From TEM images, it can be seen that the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> sample displayed rod-like morphology. The single nanorods were randomly selected, which has a diameter of 318.4 nm and length in the range of 8.15  $\mu$ m. The figure 3(e) is attributed to SAED consists of spots, which are identified as the diffraction from cubic single crystalline  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. It is in good agreement with the results obtained from the XRD pattern [24]. The HRTEM images figure 7(d) show well-defined lattice fringes of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. It is indicating the single crystallinity of the nanorods. The interplanar spacing of fringes measured to be 0.368 nm, which corresponds to the (211) planes of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. It further confirmed the single-crystalline nature of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods, and growth along the *c*-axis. Figure 6(f) showed the uniform distribution like Mn and O throughout the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods.

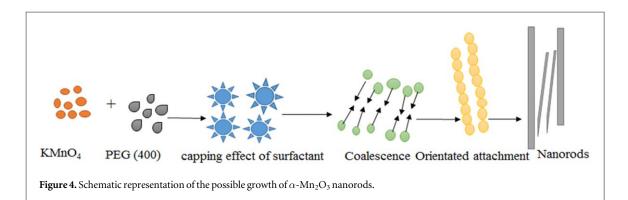


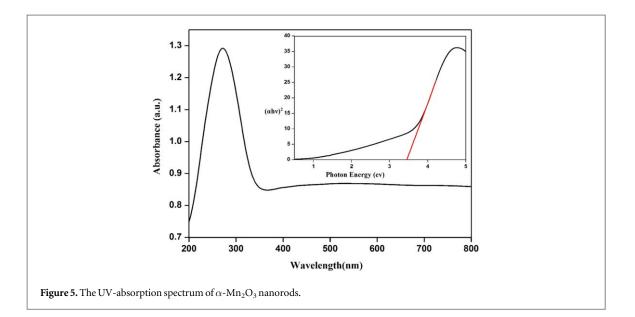
**Figure 2.** (a)–(c) SEM images of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods at different magnification. (d), (e) Elemental mapping of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. (f) EDX analysis of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods.



#### 3.3. The growth mechanism of $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods

The growth mechanism of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods illustrated in figure 4. The growth mechanism of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods fully covered by nucleation, crystal growth, coalescence, oriented attachment. A large quantity of nanorods is formed by coalescing of nanoparticles together due to oriented attachment. Oriented attachment mechanism could be described as a self-organised arrangement of neighboring nanoparticles, so that they share an identical crystallographic configuration. When the crystallographic orientation is encountered, these nanoparticles were merging together [27]. This favorable process reduces the overall surface energy due to the





elimination of energy associated with unsatisfied bonds [28]. A decrease in the length of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods could be explained by the limited lateral aggregation of nanoparticles. It may be possible to explain the sharp tip of some  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods arises because lower the surface energy and surface diffusion happened during the growth of the nanostructures [29].

#### 3.4. Optical absorption analysis

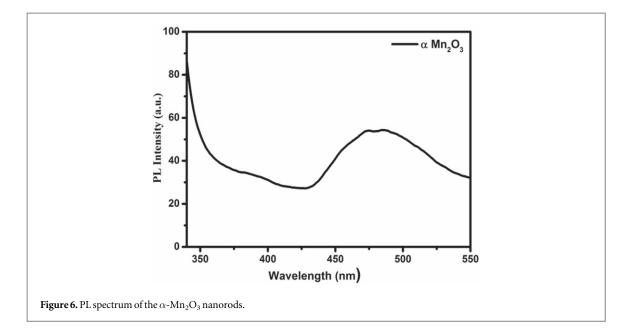
UV–visible absorption spectrum of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods were plotted in figure 5(a). A well-defined sharp and strong absorbance peak located at 300 nm due to the electron excitation from filled to the empty band was observed in  $\pi \to \pi^*$  electronic transition [14]. The energy bandgap of the sample was calculated using the Tauc expressed in a relation between the absorption coefficient ( $\alpha$ ) and the energy of the photon (h $\nu$ ) as follows,

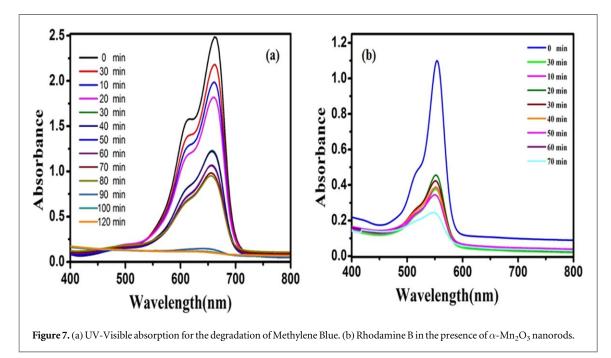
$$(\alpha h\nu)^2 = (h\nu - E_g)^n \tag{2}$$

It reveals that the obtained plotting gives to the linear portion of the curves in a certain region. The energy bandgap (Eg) was estimated using the intercept of the linear portion of the curve  $(\alpha hv)^2$  versus  $(h\nu)$  shown in the inset of figure 6(b) and was found energy bandgap was 3.4 eV. This value is reported in the early reports of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods [8]. The absorption edge and bandgap of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods showed that photocatalytic reaction under UV conditions. Especially in the light wavelength range is 300 nm in photocatalytic application. Thus, it was necessary to understand its optical property and band structure.

#### 3.5. Photoluminescence properties

The room temperature PL spectrum of the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods was examined using the Xe excitation source with excited wavelength at 300 nm is shown in figure 6. It can be seen that the PL spectrum of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods exhibits a broad blue emission band [21]. The broad blue band emission is due to the presence of oxygen vacancies that oxygen vacancies generally act as a deep defect the intrinsic point defect levels confined between the Mn 3d band, and O 2p band can result in a broader blue emission ban. In conclusion, our PL results





have evidenced the presence of vacancy in oxygen. In addition, many literature studies report oxygen vacancy was increased the photocatalytic reaction [30].

### 4. Photocatalytic dye degradation

The efficient photocatalytic activity of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods was evaluated by the degradation of methylene blue (MB) and rhodamine B (RB) under UV illumination at different irradiation times. The decolorization of dye molecules efficiency was calculated using the formula E % = (1-C/Co) \*100, where C is the residual concentration of MB and RB at different illumination time intervals. Co is the concentration of the MB and RB before illumination [31]. Figures 7(a) and (b) reports the presence of photocatalyst with (MB) and (RB) under UV lamp conditions, respectively. As the irradiation time increases, the maximum absorbance peak decreases slowly. This observation denotes that the concentration of the MB and RB decreases.

Figure 7(a) shows the complete dye degradation at 120 min for MB. Figure 7(b) rhodamine B (RB) absorbance peak was completely decreased at 70 min that indicates the total degradation of MB and RB dye and also designates the destroyed double bond of the chromophore [32]. In order to study the effect of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> photocatalyst on RB and MB, the blank without photocatalyst was also tried, which were presented in

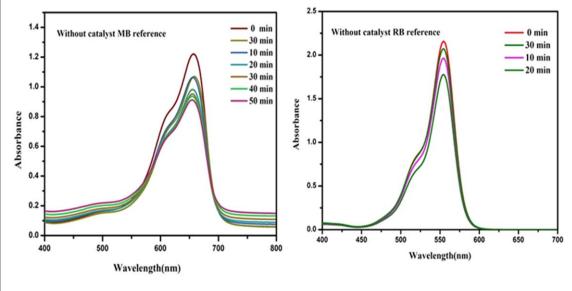
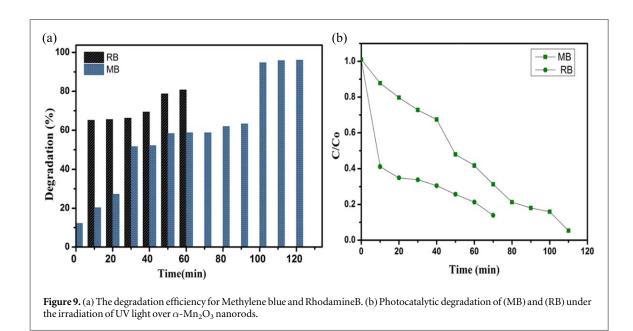


Figure 8. (a), (b) UV-Visible absorption spectra of blank Methylene Blue and Rhodamine B solution.



figures 8(a), (b), respectively. Even after 30 min of irradiation time, the absorbance peak looked slightly changed, and there was no decolorization took place. However, the significant decrease in the absorbance spectrum and the effect of decolorization were observed in 30 min of irradiation when the  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> photocatalyst added with RB and MB.

Figure 9(a) shows the % of degradation of the MB and RB dye with different photocatalyst dosage. In order to compare the efficiency of photocatalyst, the % of degradation of MB at 120 min and RB at 70 min were estimated as 95% and 80%, respectively. The values were listed in table 1, which was compared with the earlier reports in table 2. Figure 9(b) shows the decreasing concentration of the MB and RB dye versus different illumination time intervals for the samples. It depicts that the good surface states and crystallinity of single-crystalline  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods improve the photocatalytic performance [27].

#### 4.1. Decolorization of dye molecules mechanism

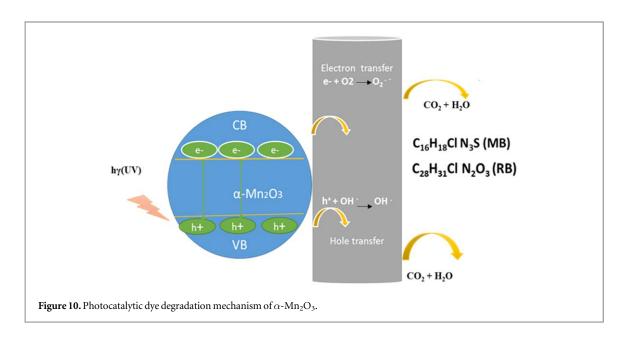
The schematic diagram in figure 10 represents the photocatalytic charge transfer taking place mechanism of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods. Photocatalytic reaction was initiated by the photon incident on the photocatalyst, where the photoelectron excited from the valence band of a photocatalyst to the conduction band [39, 40]. The photoelectron excitation process leaves behind a hole in the valence band. The net result electron and hole pairs (e<sup>-</sup>/h<sup>+</sup>) generated as a photocatalyst equation (3).

 $\label{eq:table_to_$ 

Parameters	Methylene blue	Rhodamine B	Degradation efficiency
Volume of the sample	100 ml	100 ml	
Initial concentration	2.5 mg	2.5 mg	
Weight of catalyst source	6.5 mg	4.5 mg	95% (MB)
Source	254 nm	254 nm	80% (RB)
Time taken for degradation	120 min	70 min	

<b>Table 2.</b> Comparison of $\alpha$ -Mn <sub>2</sub> O <sub>3</sub> Nanorods by absorption with	
reported literature values.	

Catalyst	Reaction time	Removal %	References
$\alpha$ -Mn <sub>2</sub> O <sub>3</sub>	120	90.2	[33]
$CeO_2/V_2O_5$	300	76.9	[34]
CeO <sub>3</sub> /CuO	300	85.7	[35]
$\alpha$ -Bi <sub>2</sub> O <sub>3</sub>	360	30	[36]
NiO	300	70.2	[37]
Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	60	80	[38]
$\alpha$ -Mn <sub>2</sub> O <sub>3</sub>	240	71	[22]
$\alpha$ -Mn <sub>2</sub> O <sub>3</sub>	120	95	Present worl
$\alpha$ -Mn <sub>2</sub> O <sub>3</sub>	70	80	Present worl



$$Mn_2O_3 + h\nu(UV) \to Mn_2O_3 + h^+(VB) + e^-(CB)$$
 (3)

While surface-bound water molecules react with the photogenerated hole ( $h^+_{VB}$ ) to produce hydroxyl radicals. The hydroxyl radicals (OH<sup>-</sup>) is a powerful oxidizing potential agent equation (5)

$$H_2 O \to H^+ + O H^- \tag{4}$$

$$OH^- + h^+(VB) \rightarrow OH^\circ(Hydroxyl radicals)$$
 (5)

Electron in the conduction band is occupied by the oxygen in order to produce anionic superoxide radicals equation (6). This reduction and oxidation process is capable of degradation of the MB and RB under UV light [10, 41].

 $O_2 + e^{-}(CB) \rightarrow O_2^{\circ -}$  (Anionic superoxide radicals) (6)

$$OH^{\circ} + MB(or)RB \rightarrow CO_{2-} + H_2O$$
(7)

$$O_2^{\circ-} + RB(or)MB \rightarrow CO_2 + H_2O$$
 (8)

In the case of one- dimensional nanostructured  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods, the photogenerated electrons can traves along the length of the crystal as the space-charge region is well constructed in the longitudinal direction

[16]. The increase in the delocalization of electrons in  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods can lead to decrease in the probability of electron-hole pair recombination. This results in the existence of a large number of charge carriers on the active sites of  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorod surface, which has better degradation efficiency as compared with spherical nanoparticles [17].

#### 5. Conclusions

In summary,  $\gamma$ -MnOOH in the monoclinic phase has been synthesized by the hydrothermal method. A cubic bixbyite  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> nanorods were obtained by calcined  $\gamma$ -MnOOH at 600 °C for 4 h. XRD, SEM, TEM, and HRTEM showed the formation of cubic phase structure and well-defined large quality of rod-like morphology with a diameter of 318.4 nm length 8.15  $\mu$ m, and growth mechanism of nanorods  $\alpha$ -Mn<sub>2</sub>O<sub>3</sub> was discussed. The decolorization of dye molecules performance for the methylene blue and rhodamine B dye solution under UV irradiation for 120 min and 70 min. The photocatalytic dye degradation efficiency was estimated as MB (95%), RB (80%). The results show that the degradation efficiency also depended on the photocatalyst dosage.

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